

## ARTICLE

Received 7 Oct 2013 | Accepted 15 Jan 2014 | Published 7 Feb 2014

DOI: 10.1038/ncomms4266

# Fractal design concepts for stretchable electronics

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Stretchable electronics provide a foundation for applications that exceed the scope of conventional wafer and circuit board technologies due to their unique capacity to integrate with soft materials and curvilinear surfaces. The range of possibilities is predicated on the development of device architectures that simultaneously offer advanced electronic function and compliant mechanics. Here we report that thin films of hard electronic materials patterned in deterministic fractal motifs and bonded to elastomers enable unusual mechanics with important implications in stretchable device design. In particular, we demonstrate the utility of Peano, Greek cross, Vicsek and other fractal constructs to yield space-filling structures of electronic materials, including monocrystalline silicon, for electrophysiological sensors, precision monitors and actuators, and radio frequency antennas. These devices support conformal mounting on the skin and have unique properties such as invisibility under magnetic resonance imaging. The results suggest that fractal-based layouts represent important strategies for hard-soft materials integration.

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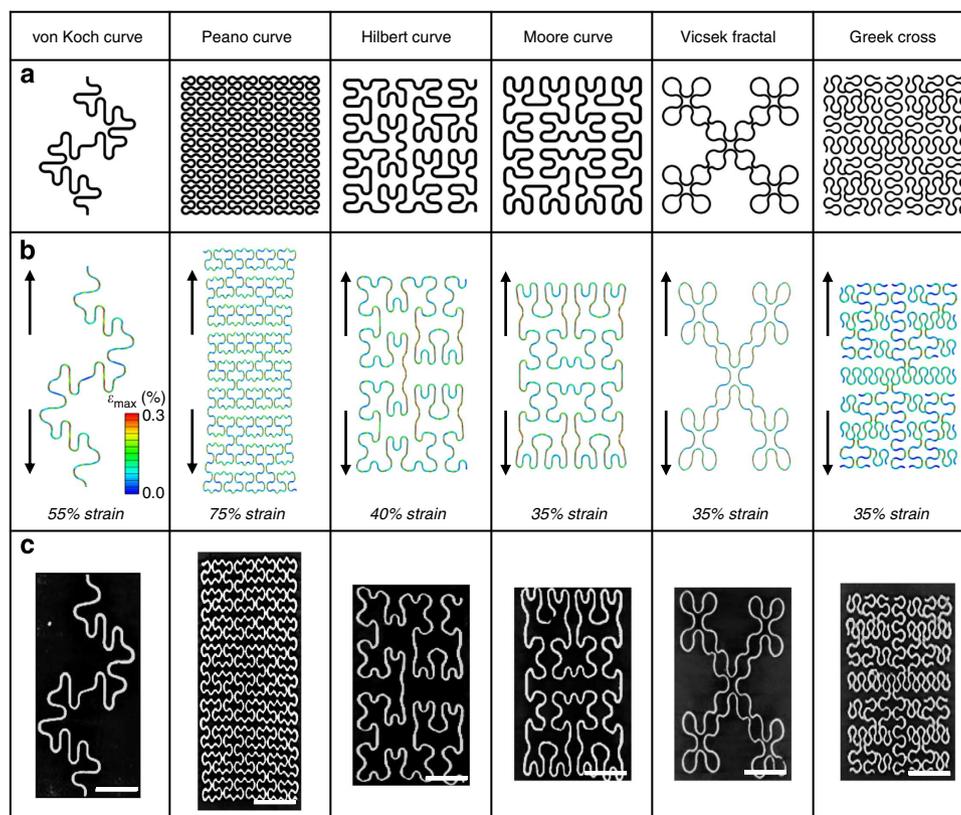
The field of stretchable electronics is of growing interest<sup>1,2</sup>, motivated both by fundamental considerations in material science and by application spaces in areas such as biomedicine<sup>3–6</sup>. A core challenge is in achieving high-performance electronic functionality with systems that offer elastic, low modulus responses to large strain deformations. Two of the most successful approaches to this problem exploit advanced composites, in a generalized sense. The first involves dispersing conducting or semiconducting nanomaterials (that is, nanowires, nanotubes or graphene) into elastomeric matrices<sup>7–11</sup>. Here, three-dimensional (3D) structures, including distributed networks of contacts, form spontaneously, but with limited direct control over key variables other than the overall loading fraction. The second exploits alternative classes of composites, created deterministically by using thin layers of electronic materials lithographically defined into two dimensional (2D) filamentary mesh layouts<sup>12–14</sup>. Advantages of this approach include the ability to co-integrate multiple high-performance material platforms with high spatial resolution and engineering control, in ways that also allow proper electrical contacts, both internal and external to the system<sup>15</sup>. An essential aim of both 3D random and 2D deterministic composites in stretchable electronics is to combine high loading of a hard component, for use in active devices, with overall mechanics that is dominated by a soft matrix, for stretchable response. This goal is opposite to that associated with traditional composites engineering and therefore demands alternative approaches.

Here, we show that concepts in fractal geometry, which are known to determine behaviours in traditional 3D

networks<sup>16–20</sup> and which are pervasive in biological systems including the human body<sup>21,22</sup>, can be successfully exploited in 2D deterministic systems, with important functional consequences in advanced stretchable electronics. Fractal-based structures can be described by self-similarity: subdivision into small sections yields pieces with geometries that resemble the whole. Compared with previously explored networks of periodic serpentine shapes, fractal designs can be engineered to accommodate enhanced elastic strain along a selected dimension and to support biaxial, radial and other deformation modes. In addition, the choices of topologies span a rich range, from lines to loops, capable of tailoring to specific electronic applications through integration and interdigitation of multiple structures. Figure 1 presents six representative examples, from lines (Koch, Peano, Hilbert) to loops (Moore, Vicsek) and branch-like meshes (Greek cross). The results illustrate the diversity of possibilities, through both the finite element method (FEM) and experimental demonstration. The approximate fractal dimensions in these finite-iterative curves range from 1.5 to 2. The elastic tensile strains achieved with these structures indicate that they are suitable for use in various stretchable devices, including the epidermal electronic platform<sup>23</sup>, with key advantages over previously described layouts.

## Results

**Mechanics and electronics with Peano-based geometries.** The Peano curve<sup>24</sup> provides a model system for examining the detailed mechanics of fractal-based motifs. Layouts using these or



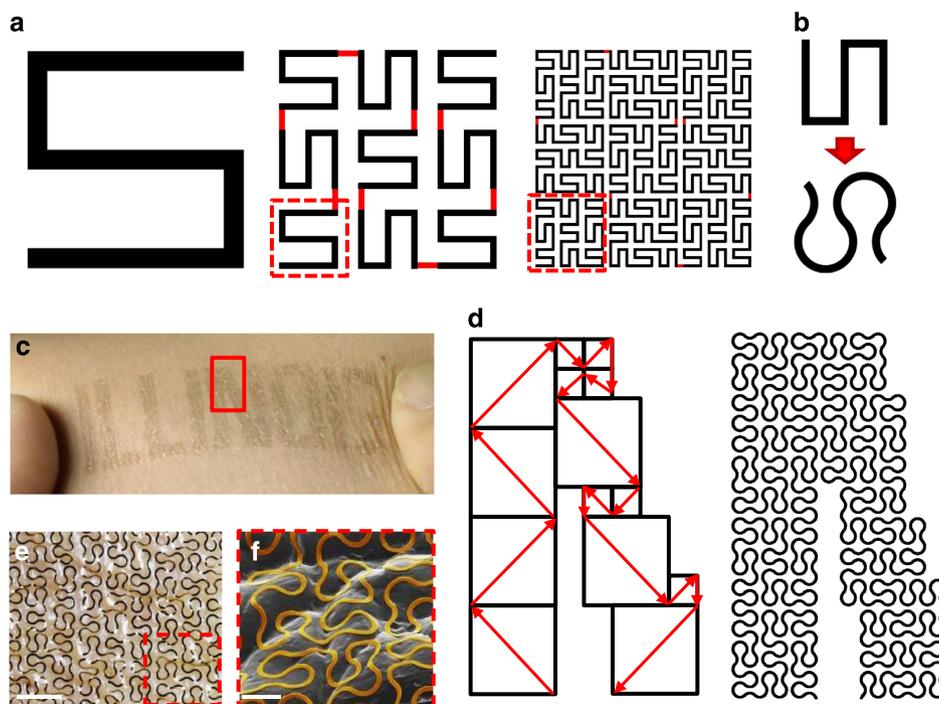
**Figure 1 | Representative fractal-inspired layouts for hard-soft materials integration.** (a) Six different patterns of metal wires fully bonded to elastomer substrates demonstrate the application of deterministic fractal designs as general layouts for stretchable electronics. These patterns include line, loop and branch-like geometries, and they are applicable to a broad range of active and passive electronic components. In all cases, arc sections replace the sharp corners from the mathematically defined fractal layouts to improve the elastic mechanics. (b) FEM images of each structure under elastic tensile strain and (c) their corresponding experimental MicroXCT images demonstrate the elastic mechanics. The wires consist of layers of gold (300 nm) sandwiched by polyimide (1.2  $\mu\text{m}$ ) and mounted on an elastomeric substrate (0.5 mm). Scale bars, 2 mm.

other designs follow an iterative approach to create the  $N$ th Peano curve, nine copies of the  $(N - 1)$ th curve are linked together into a single line. An illustration of the first three iterations of a Peano curve appears in Fig. 2a. A physical interpretation that captures the underlying mechanics follows by considering the first iterative curve geometry as a section of a one-dimensional spring. Because of self-similarity, higher order Peano curves contain spring-like motifs at multiple length scales (Supplementary Fig. 1). Furthermore, each of these motifs can be oriented vertically ( $y$ -axis) or horizontally ( $x$ -axis). As such, the Peano geometry provides a design metric for systematically ‘folding’ a straight line into a compact and stretchable layout. To further enhance the mechanics of these wires, arc sections replace the sharp corners in the mathematically defined Peano curves (Fig. 2b). Generally, the elastic mechanics of horseshoe serpentine structures improve with increasing arc section angle (Supplementary Table 1). Although the Peano curves featured in Fig. 2a fill square areas, those of different iterative orders can be linked together to fill spaces of arbitrary shape. As a demonstration, Fig. 2c–f show the word ‘ILLINOIS’ formed with space filling wires, in which each letter consists of a combination of first and second order Peano curves. The figure includes a detailed view of a section of the ‘N’. Electrodes can be constructed in this manner to match, for example, particular features on the body.

The Peano curves in Fig. 2a represent only one set of variations; there exist many others, because the  $(N - 1)$ th unit cells of any  $N$ th order curve can be oriented either vertically or horizontally (Supplementary Fig. 2). The second order curve has 272 unique layouts alone, each with distinct mechanical properties. Numerical simulations of five different second order layouts

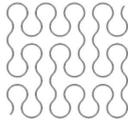
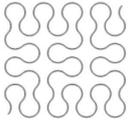
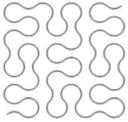
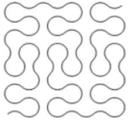
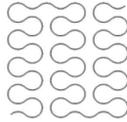
consisting of gold wires bonded on an elastomer probe the relationship between mechanics and layout. The layouts contain unit cells with orientations ranging from all vertical to all horizontal (Table 1). The maximum principal strain criterion defines the maximum elastic stretchability as the onset of plastic yield, consistent with established parameters of the constituent materials<sup>25</sup>. Table 1 summarizes this quantity, calculated for uniaxial deformation along the  $x$ - and  $y$ -axis. The results indicate that Peano layouts with unit cells all oriented in the same way maximize the uniaxial stretchability along the unit cell direction. The ‘half-and-half’ Peano layout, which contains unit cells with alternating orientations, balances the maximum strain supported along the  $x$ - and  $y$ -axis at 16% and 13%, respectively. Such properties are well suited for devices that stretch along both axes. Adjustments to the total unit cell size can enhance the mechanics of these structures (Supplementary Table 2). The third order half-and-half layouts yield stretchabilities along the  $x$ - and  $y$ -axis of 32% and 28%, respectively (Supplementary Fig. 3). This improvement with iteration order is due to the combined effects of geometric scaling of the arc sections, increase in the length of the wire and addition of higher order spring-like motifs. The stretchability here is well above 20%, which is the general upper limit for the elastic mechanics of skin<sup>23</sup>.

Experimental structures consisting of second and third order half-and-half Peano layouts of metallic wires that are clad with polyimide and fully bonded to an elastomeric membrane compare well with the numerical analysis (See Methods for details). The polyimide cladding retards strain localization in the metal<sup>26</sup>, thereby enhancing sample yield and consistency in the mechanical tests. Three samples are tested for each fractal



**Figure 2 | Implementation of Peano curves for stretchable electronics.** (a) Illustration of three iterations of a two-dimensional Peano curve. Nine versions of the  $(N-1)$ th curve (dotted red box) connect together with the solid red lines to construct the  $N$ th curve. (b) Arc sections replace the sharp bends to enhance the mechanics. (c) Image of metal wires with Peano layouts, with an overall geometry that spell out the characters in ‘ILLINOIS’, mounted on skin. Here, each letter consists of a series of first and second order Peano curves. (d) A detailed section of the  $N$  (solid red box) from (c) in block diagram form (small blocks are first order curves, large blocks are second order curves) and as wires. (e) Optical and (f) scanning electron microscopy images of third order Peano-based wires on skin and a skin-replica (colorized metal wires), respectively, showing the conformal contact of the wires on the substrate. Scale bar in (e), 2 mm. Scale bar in (f), 500  $\mu\text{m}$ .

**Table 1 | Elastic mechanics of five different Peano-based wire structures.**

	All-Horizontal	Mostly-Horizontal	Half-and-Half	Mostly-Vertical	All-Vertical
Config.					
x-axis	31%	16%	16%	11%	9%
y-axis	9%	12%	13%	14%	33%

Calculated stretchability of metal wires mounted on an elastomer in five different second order Peano layouts, given a maximum principal strain criterion of 0.3% in any section of the wires. The layouts range from 'all-horizontal' (subunits are all oriented along the x-axis) to 'all-vertical' (subunits are all oriented along the y-axis). The strain criterion defines the transition from elastic to plastic deformation in the wires.

dimension and axis of stretching. Mechanical characterization involves measurements of resistance changes to a precision of  $\sim 0.01$  ohms with a four-point probe technique performed during uniaxial tensile stretching. In the elastic regime, the wire resistance before and after cycling between states of no strain and increasing levels of maximum strain (the difference defined as the 'differential resistance') does not change. At the onset of plastic deformation, the wire undergoes a resistance-dependent geometric change, leading to a small but measurable differential resistance. Traditional approaches to characterizing plasticity<sup>27,28</sup> based on measurements of stress-strain response with a tensometer are not suitable because the mechanics of the elastomeric substrates in these systems dominate the response, by design.

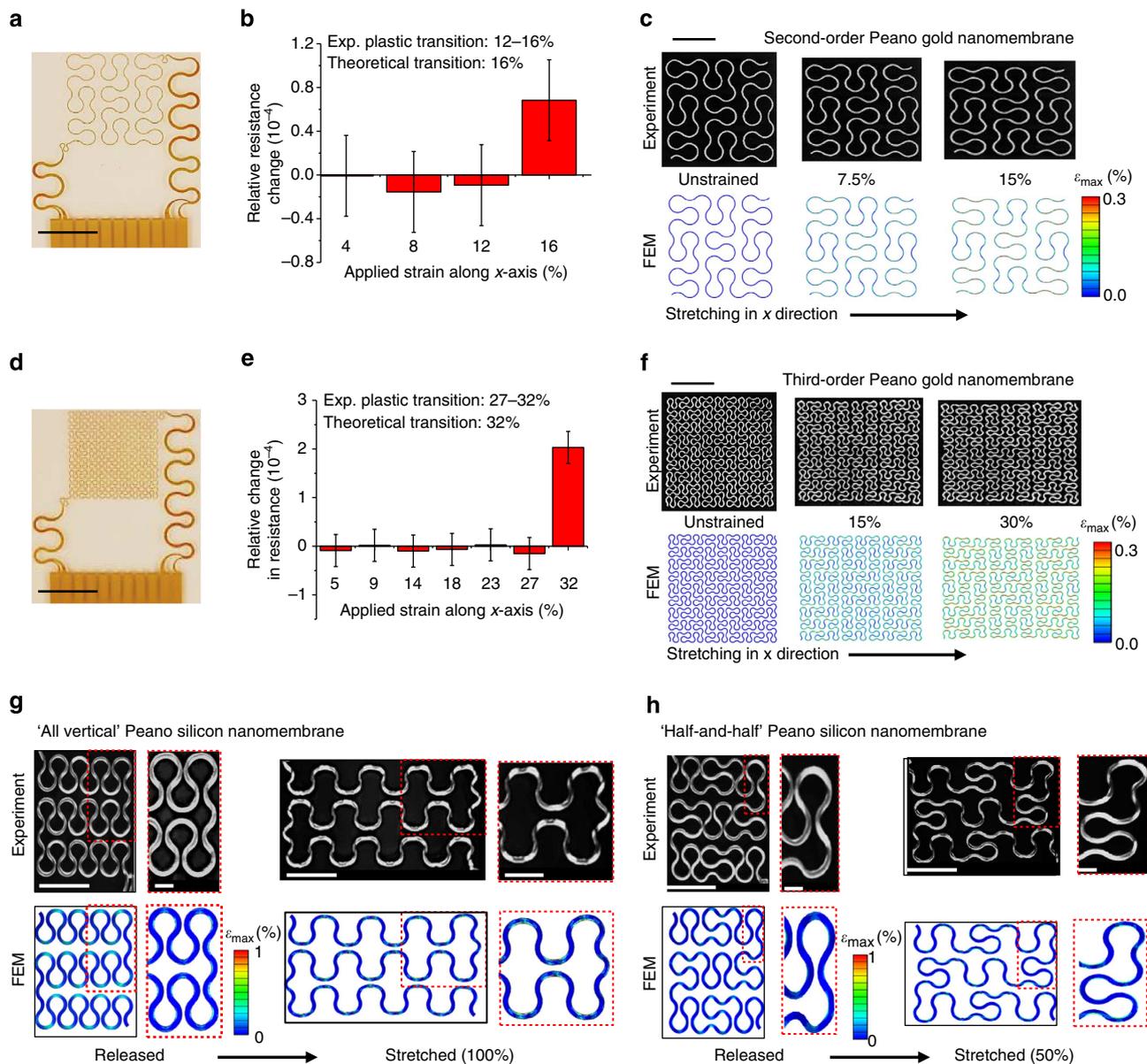
Figure 3a–f and Supplementary Fig. 3 summarize the results of these studies. The differential resistances measured from individual representative devices appear together in each plot. The error bars in the second and third order Peano structures have magnitudes of  $3.7e-5$  and  $3.3e-5$ , respectively; Supplementary Note 3 discusses the sources of these errors. Simultaneous measurements of the local temperature account for and calibrate out changes in resistance due to background temperature fluctuations (See Supplementary Note 2). The measurements show that for the second order Peano structures stretched along the  $x$ - and  $y$ -axis, the transition from elastic to plastic deformation occurs in the range of 16–20% and 12–16%, respectively. The third order Peano structures undergo an elastic-to-plastic transition in the range of 27–32% for stretching along both  $x$ - and  $y$ -axes. These values of uniaxial stretchability are consistent with the numerical analysis. Calculated FEM images of strain localized in the stretched structures (Fig. 3c,d) display the positions of the mechanical 'hot spots' in the samples, where strains are particularly large and where failure occurs. Further enhancements in device mechanics are possible by optimizing the detailed geometry of these hot spots.

Fractal-based structures bonded to pre-strained elastomers enable higher levels of elastic deformation<sup>29,30</sup>. A second order all-vertical Peano structure fully bonded onto an elastomeric substrate with 40% pre-strain demonstrates the concept. The differential resistances for different levels of maximum applied strain appear in Supplementary Fig. 4. Here, the transition from elastic to plastic deformation occurs when the substrate is strained in the range of 60–70%, which is significantly higher than that in samples prepared without pre-strain. Schemes that use pre-strain can be extended to biaxial, radial or other types of deformation.

This concept of enhancing mechanics through the use of pre-strain is general to a broad range of materials, including

semiconductors. Thin films of single crystalline silicon nanomembranes (Si NM) with two different second order Peano layouts and bonded onto 40% pre-strained elastomeric substrates provide a proof of concept. In the pre-strained state, the calculated compressive stresses in the membrane are well within the regime of elastic deformation for silicon. Figure 3g,h shows microscale X-ray coherent tomography (microXCT, see Supplementary Note 1) images of the samples, along with corresponding FEM results. The findings indicate that the all-vertical and half-and-half structures can be elastically strained by 105% and 63%, respectively, given a maximum principal strain of 1% in the silicon<sup>31</sup>. The fracture points measured electrically from half-and-half structures (Supplementary Fig. 5) are consistent with the FEM results. Both the microXCT and FEM images reveal microscale buckling; this type of deformation mode improves the ability of the Si NMs to dissipate stress. Such behaviour persists only for a limited range of NM thicknesses. FEM simulations show that, with increasing membrane thickness, the NMs transition from a regime of wrinkling to microscale buckling and finally to global buckling; furthermore, the microscale buckling regime yields maximal elastic mechanics (Supplementary Fig. 6). As such, the optimization of the elastic properties of hard-soft systems requires careful attention to micro-mechanics.

**Fractal-based epidermal electronics.** The Peano and other fractal layouts have utility for various applications. One is in skin-mounted sensors and actuators such as temperature sensors, heaters and electrodes (Supplementary Fig. 7). Electrodes measure electrophysiological processes in the brain (electroencephalograms (EEGs))<sup>23</sup>, heart (electrocardiograms (ECGs)) and muscle (electromyograms (EMGs))<sup>32,33</sup>. To minimize impedance between the electrode and skin and to optimize the measured signal-to-noise, electrodes require both conformal skin contact and high areal coverage<sup>32,33</sup>. Electrodes that interface directly with neurons additionally benefit from having large perimeters within an area<sup>34–36</sup>. These needs can be addressed effectively using a variant of the Greek cross fractal, which consists of a hierarchy of cross structures that fills space in two dimensions (Supplementary Fig. 8). This design embeds a high level of connectivity, which minimizes the resistance between any two points. In addition, defects in the structure, such as line breaks, have a reduced probability for significantly affecting device performance, which is desirable for robust, long-term health monitoring in practice<sup>33</sup>. These geometries can be further designed to eliminate closed loops (Supplementary Fig. 8), such that the edges of the electrode wire layout form a single continuous line. A multifunctional device that incorporates a



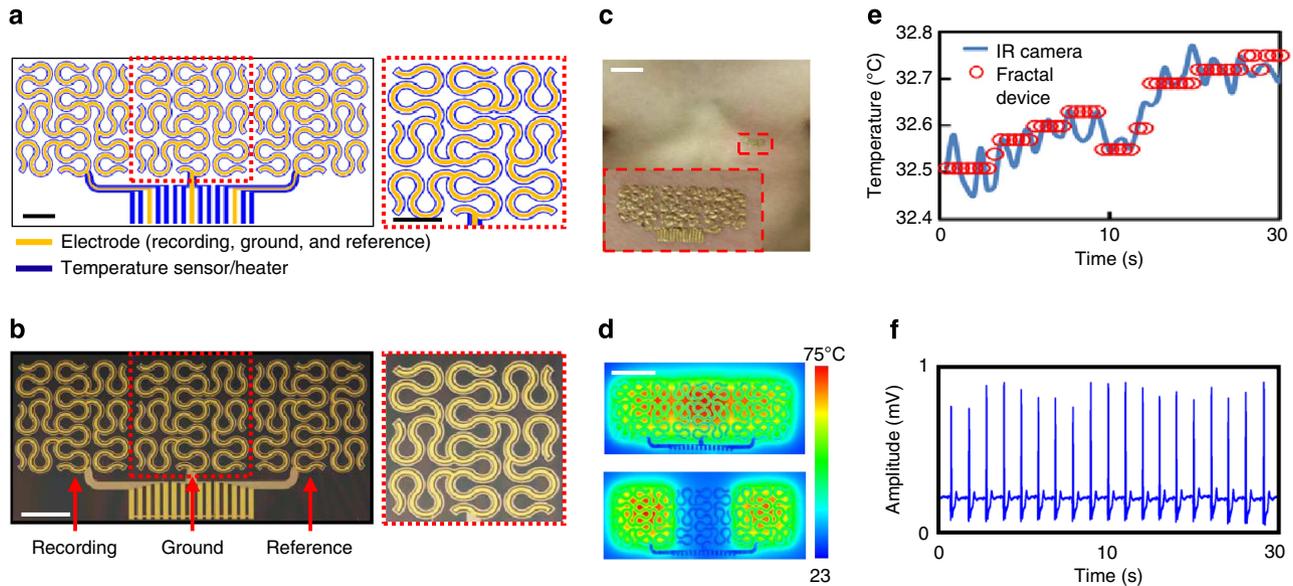
**Figure 3 | Mechanics of metal and semiconductor fractal-based structures.** (a,d) Optical images of metal wires patterned with second and third order half-and-half Peano layouts. The samples connect to a stretchable set of four wires used for four-point probing. Scale bar, 5 mm. (b,e) Plot of differential resistances measured at increasing levels of maximum applied strain. The onset of plastic deformation corresponds to the strain at which the differential resistance is non-zero. (c,f) A comparison between experimental optical images and FEM-constructed images of the second and third order structures for different levels of stretching. The FEM images display localized areas of high strain in the stretched structures. Scale bar, 3 mm. (g,h) MicroXCT and FEM images of Si NMs patterned into Peano layouts and fully bonded to a 40% pre-stretched elastomeric substrate. Upon release and uniaxial stretching, both structures dissipate the mechanical stress via microscale buckles, which is indicative of a mechanical regime well suited for high levels of elastic deformation. Scale bar, 1 mm.

resistive temperature sensing/heating element with an integrated electrode (recording, ground and reference components together) exploits this feature<sup>30</sup> (Fig. 4a). The temperature sensor<sup>37</sup> consists of an insulated line of metal running along the edges of the Greek cross electrode. Figure 4b–f shows a representative device and its operation under different modes (heating, temperature sensing, ECG measurements). The impedances and signal-to-noise measured with these dry electrodes compare favourably to that of conventional gel-based electrodes (Supplementary Fig. 9), indicating that they are suitable for high quality, skin-mounted, electrophysiological measurements. The precision of the temperature measurement ( $\sim 20$  mK) compares well with that of an infrared camera.

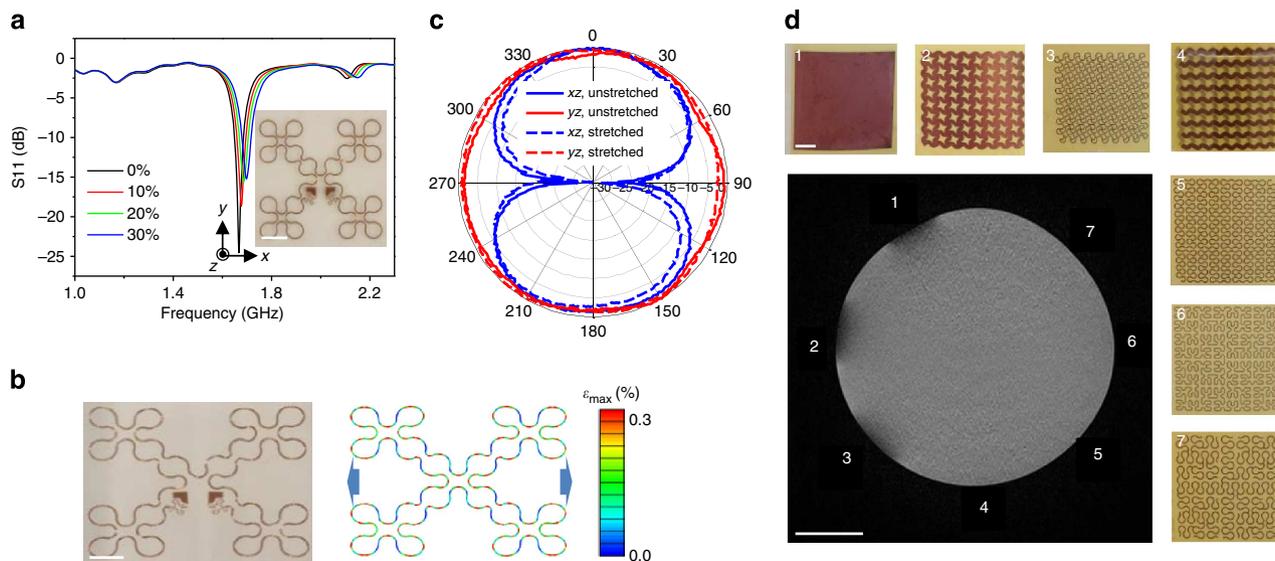
**Radio-frequency devices with fractal layouts.** Stretchable radio frequency antennas, which are broadly applicable as mechanically-tunable electronic components<sup>38,39</sup>, are another class of devices that benefit from concepts in fractal design. Fractal antennas have been a topic of interest because they can support multiband operation in spatial scales that are compact relative to the resonant wavelength<sup>40–42</sup>. Appropriate choices of fractal layout offer not only this characteristic but also attractive elastic mechanics when mounted on stretchable substrates. A Vicsek curve loop antenna (Supplementary Fig. 10), in which arc sections replace sharp bends, provides a model system. The antenna consists of copper traces ( $3\ \mu\text{m}$  thick) laminated with polyimide, and bonded onto a 1 mm thick elastomeric substrate.

The copper thickness is comparable to the skin depth of copper ( $\sim 2 \mu\text{m}$ ) at gigahertz frequencies. The return loss spectrum for the unstrained antenna displays a fundamental mode near 1.7 GHz (Fig. 5a) with an impedance of 42 ohms at resonance. The total length of the antenna at resonance is approximately  $\lambda_0/6$ , where  $\lambda_0$  is the free space wavelength, reflecting the compact nature of this particular fractal layout. As the device is strained, its

fundamental frequency and input impedance slightly shift. Far-field measurements in an anechoic chamber provide additional information; data for the fundamental mode at 0 and 30% strain (Fig. 5c) display a clear dipole-like pattern. The realized gain for both the unstrained and stretched devices ranges from  $-2$  to 0 dB, which is slightly less than that for an ideal dipole due to ohmic losses in the thin copper wires. Simulations of the return



**Figure 4 | Fractal-based epidermal devices.** (a) Schematic image of a multifunctional device based on a Greek cross fractal design, with corresponding images of the device on a wafer (b) and mounted on skin (c). The device has multiple operation modes, including those for (d) Joule heating, by driving currents through the insulated heater wires, (e) temperature sensing via four-point probing, and (f) ECG measurements, in which the patch integrates the recording, reference and ground electrodes. Scale bars in (a) and (b), 2 mm. Scale bars in (c) and (d), 5 mm.



**Figure 5 | RF properties of stretchable fractal structures.** (a) Return loss parameters of a box fractal antenna under different amounts of tensile strain. Inset: optical image of an unstrained antenna fully bonded onto an elastomer. Scale bar, 4 mm. (b) Optical image of a box fractal antenna under 30% tensile strain and a corresponding FEM-constructed image of the same structure. Scale bar, 4 mm. (c) Far-field profiles of the antenna under no strain and 30% strain. (d) Cross-sectional MRI image of different copper samples (labelled 1-7) attached to a bottle of water. The solid and mesh samples display shadowing in the MRI image, while the fractal-based samples display no such shadowing. The sample layouts all are  $1 \times 1$  inches and are (1) solid square, (2) thick mesh, (3) thin mesh, (4) horizontal and vertical stripes separated by a spacer, (5) Peano curve, (6) Hilbert curve, (7) Serpentine Greek cross. Scale bar, 5 mm.

losses and far field profiles are consistent with the experiments (Supplementary Fig. 11, Supplementary Note 4).

Another application that can benefit from the favourable RF properties and mechanics of fractal patterns is in electrode structures that are compatible with magnetic resonance imaging (MRI). Copper foil samples mounted onto a cylindrical phantom and scanned in a 3-Tesla Trio head scanner (Siemens Instruments) provide a demonstration. For purposes of comparison, the samples include three types of fractals, along with an unpatterned sheet, two variants of serpentine meshes, and superimposed vertical and horizontal lines. The fill fraction of the meshes and the fractal patterns are approximately the same ( $\sim 25\%$ ). For the magnetic field strength used here, the resonant RF pulse frequencies are  $\sim 123$  MHz. Figure 5d displays an axial cross-sectional scan obtained using a spin echo imaging sequence (parameters are in Supplementary Note 5). The water-based phantom appears white due to its high hydrogen concentration. The MRI image shows clear shadows in the vicinity of the unpatterned film and the mesh samples. Conversely, the fractal samples exhibit no noticeable distortion or shadowing. Magnetostatic coupling between RF radiation and the samples, which yields circulating currents and associated heat dissipation and signal loss, explains these differences. The meshes couple to RF radiation because they consist of highly interconnected closed loops of metal; in contrast, the fractals do not contain closed loops, do not couple to RF radiation and are invisible in the MRI. This analysis suggests that fractal-based designs provide routes to MRI-compatible skin-mounted or implanted electronic devices.

In summary, fractal-based layouts create new design opportunities in stretchable electronics, including a broad range of devices suitable for biomedical systems. One of the challenges here is evaluating the mechanical properties of these composite materials and rigorously identifying their elastic and plastic mechanics. With the combination of high precision electro-mechanical measurements and three-dimensional FEM simulations, the fundamental mechanical responses and their dependence on geometry can be understood and exploited for specific deformation modes. This study suggests a general relationship between fractal layouts and mechanics that is broadly applicable to stretchable materials engineering.

## Methods

**Epidermal device fabrication.** Fabrication of metal-based devices, including the temperature sensors and heaters, involves first spin-coating and curing liquid polyimide (PI) on a UV-treated PDMS-coated Si wafer. Electron beam evaporation yields layers of chromium (4 nm) and gold (300 nm). Following metal patterning, etching and the application of a second layer of polyimide, a photoresist mask and oxygen plasma etch define the polyimide-encapsulated device. All of the wires are 70  $\mu\text{m}$  wide and the polyimide layers are each 1.2  $\mu\text{m}$  thick. Water-soluble tape (3M) retrieves the gold-polyimide structures, which covalently attach to an elastomer (0.5 mm thick, Ecoflex or Solaris, Smooth-on Inc.) using an evaporated chromium-silica bonding layer. UV radiation pretreatment of the elastomer promotes covalent bonding between the silica and elastomer. An ACF cable (Elform, USA) attached to the bonding pads of the device enable accessible electrical addressing. The same process, with an additional polyimide etch step, applies to the open metal-based fractal electrodes for electrophysiological sensing.

An SOI wafer consisting of 260 nm thick p-doped silicon on a 1  $\mu\text{m}$  thick silicon dioxide layer is the starting point for the Si NM samples. To detach the Si NMs from the handle wafer, an HF wet etch dissolves the underlying silicon dioxide through an array of 3  $\mu\text{m}$  wide holes defined in the membranes. A PDMS stamp transfers the membranes onto a polyimide film, and a photoresist mask and dry etching process define the membranes into various Peano layouts. Gold wires electrically address the devices, and the same transfer steps described above finalize the devices.

**Antenna fabrication.** The starting material is copper foil (3  $\mu\text{m}$ ) on a copper carrier (35  $\mu\text{m}$ ; Olin Brass). Polyimide spun-cast and cured onto the foil yield foil-polyimide laminates, which mount onto a PDMS-coated substrate and enable copper carrier removal. A photoresist mask, wet copper etch and oxygen plasma dry etch pattern the antenna.

**Fractal-based metal wire simulations.** FEM yields the theoretical deformation, elastic-to-plastic transition and fracture of the structures. For simulations presented in Figs 1 and 3 and Table 1, the elastomeric substrates employ an 8-node, hexahedral brick solid element C3D8R in the FEM programme, and the thin wires of PI/Au/PI-layered geometry employ a quadrilateral shell element S4R with the section of composite layout. All of the wires are 70  $\mu\text{m}$  wide and consist of a 300 nm thick gold layer sandwiched by 1.2  $\mu\text{m}$  thick polyimide layers on each side. The total pattern areas are 7 by 7 mm and fully bond to a 0.5 mm thick elastomer with a modulus of 50 kPa. The solid elements bond together physically and therefore share nodes with its adhered shell elements. An ideal elastic-plastic constitutive relation with a Young's modulus of 78 GPa, Poisson's ratio of 0.44, yield strain of 0.3% and fracture strain of 5% describe the mechanical behaviour of Au. The elastic-plastic transition is set when the maximum strain of half the width of one section is beyond the yield strain of 0.3%.

**Electrode and temperature sensor testing.** The Greek cross electrodes record ECG signals from the torso. Scotch tape exfoliates the stratum corneum and an alcohol swab removes dirt or particles to reduce effects of hydration and surface impedance. Here, the ground electrode, located between the measurement and reference electrodes ( $\sim 7$  mm apart at center-to-center distance), defines the common zero potential. Measured signals transmit wirelessly to a receiver, and commercial software using a 60 Hz notch filter and low-pass Butterworth filters (BioRadio 150, Cleveland Medical Devices, USA) completes the analysis. The fractal temperature sensors operate using the same four-point probe technique described in the mechanical testing section. An IR camera and hot plate yields  $dV/dT$  used to calibrate the devices. The devices mount directly onto the skin with no elastomeric backing layer with a spray bandage technique.

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### Acknowledgements

J.A.F. thanks Michael Harland and Roger Smith for machining work, Miguel Moscoso for sample preparation, and Chad Webb and Andrew Bonifas for useful discussions. W.H.Y. thanks Ilyoun Na, Dongsup Lee, Philip Won and Ohjin Kwon for help with material and device preparation, imaging processing and data analysis. H.C. acknowledges the Howard Hughes Medical Institute International Student Research Fellowship for support. Research is supported by the US Department of Energy, Office of Basic Energy Sciences, Division of Materials Sciences and Engineering under Award # DE-FG02-07ER46471, through the Frederick Seitz Materials Research Laboratory at the University of Illinois at Urbana-Champaign. T.P.C. is supported by ONR W911QY-12-C-0090 and ARO-62793-RT-REP. Financial support from the NSF is acknowledged for the theoretical analysis.

### Disclaimer

All experiments on human skins were conducted under approval from Institutional Review Board at the University of Illinois at Urbana-Champaign (protocol number: 13261). There are two subjects (age: 21 ~ 32, all males), co-authors in the paper. Research was carried out with informed signed consents from the subjects.

### Author contributions

J.A.F. and W.-H.Y. designed the project; J.A.F., Y.H., W.L., S.-Y.J., L.F., D.G. and R.L. carried out experiments; Y.S., Y.Z., Z.L., H.C. and M.B. carried out simulations; and T.C., Y.H. and J.A.R. supervised the project. All of the co-authors contributed to discussion and analysis of the data.

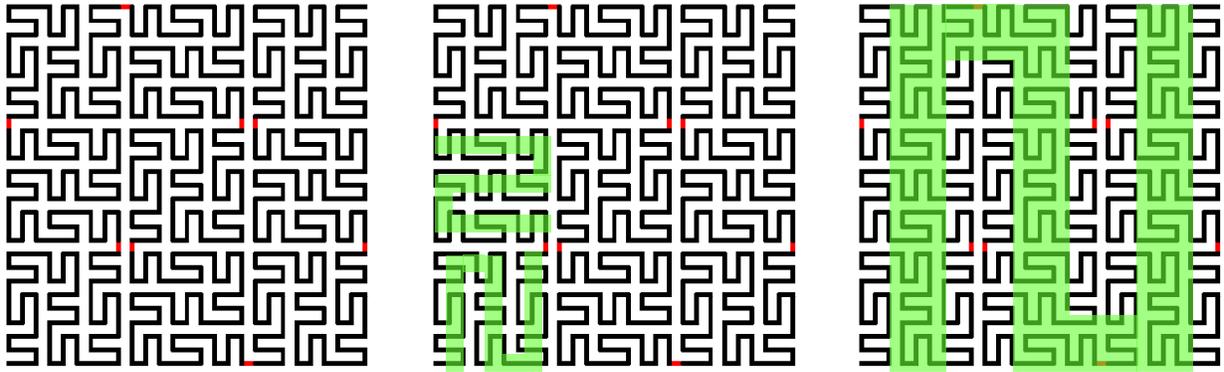
### Additional information

**Supplementary Information** accompanies this paper at <http://www.nature.com/naturecommunications>

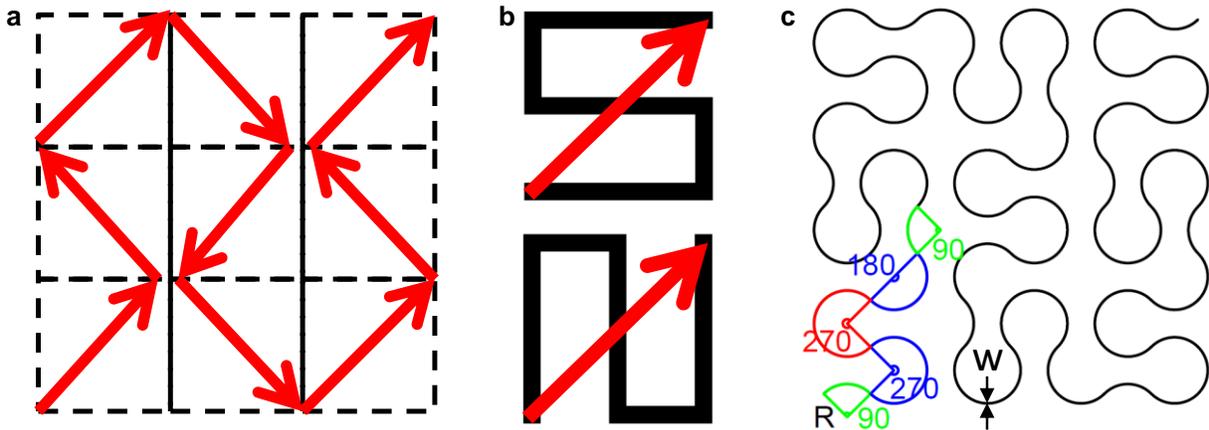
**Competing financial interests:** The authors declare no competing financial interests.

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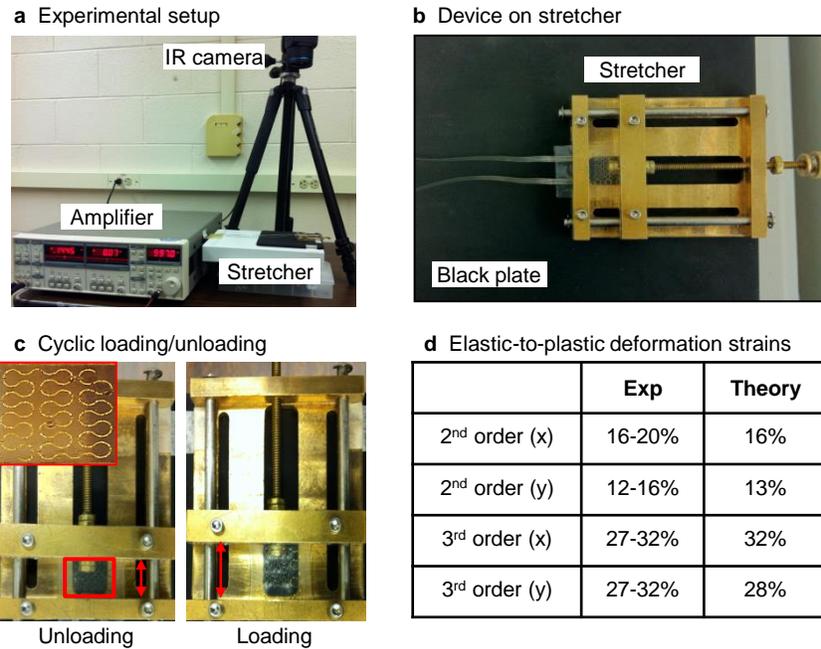
**How to cite this article:** Fan, J. A. *et al.* Fractal design concepts for stretchable electronics. *Nat. Commun.* 5:3266 doi: 10.1038/ncomms4266 (2014).



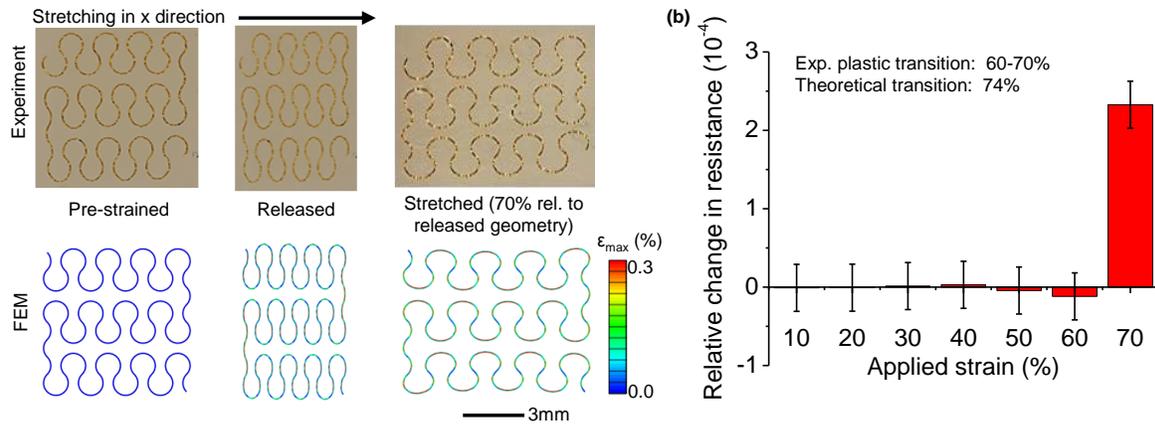
**Supplementary Figure 1.** The third order Peano curve, as depicted in Figure 1a, contains spring-like motifs at small (left), medium (middle, green highlight), and large (right, green highlight) length scales. Additionally, the spring-like motifs can orient along the x- or y-axes, as illustrated by the two medium springs in the middle plot.



**Supplementary Figure 2.** (a) Nine versions of the (N-1)th sub-unit cell link together into a continuous line to construct the Nth iteration is constructed by linking together. The arrows denote the starting and ending point of each of the sub-unit cells. Each of these nine subunits can orient either vertically or horizontally without changing its starting and ending point, as demonstrated with the first order iterative subunit (b). As such, there exist many variations of the Peano curve (272 different second order Peano curves alone). (c) The simulated and experimental structures consist of a series of arc sections with solid angles of 90, 180, and 270 degrees. The intersection points between arc sections coincide with the mathematically-defined Peano curve, which constrains their position. As such, the radii of curvature (R), wire width (w), and wire thickness (t) define the absolute geometry for a given layout type and fractal dimension. In Figure 3, the experimental dimensions are as follows: the 2<sup>nd</sup> order gold wires have R = 620  $\mu\text{m}$  and w = 70  $\mu\text{m}$ ; the 3<sup>rd</sup> order gold wires have R = 190  $\mu\text{m}$  and w = 70  $\mu\text{m}$ ; and the silicon nanomembranes have R = 190  $\mu\text{m}$  and w = 70  $\mu\text{m}$ .

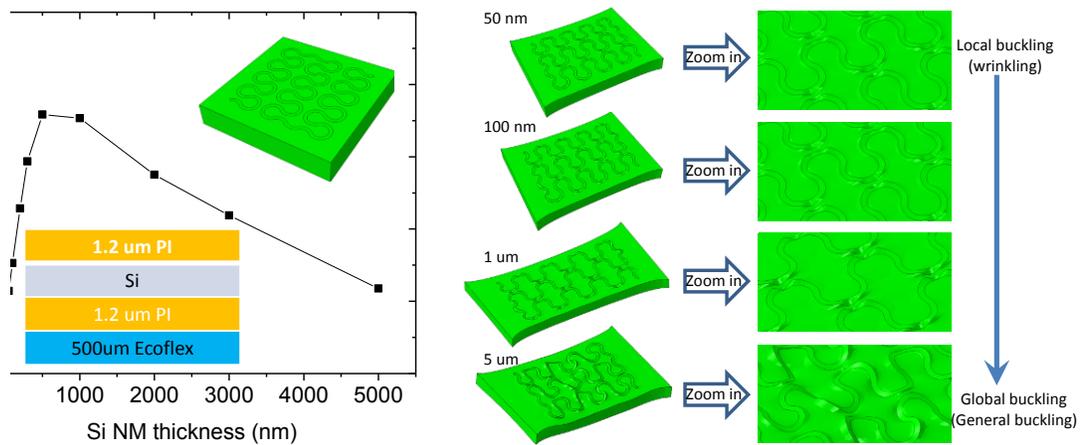


**Supplementary Figure 3.** (a) Experimental setup for device calibration and testing. To measure the device resistance via four point probe, we supply 1mA of constant current to the device with one pair of wires and measure the voltage drop in the device with the second pair of wires using a lock-in amplifier. (b) Detailed picture of the fractal-based sample bonded to ACF cables and mounted on the uniaxial stretcher. (c) Detailed picture of the prestrained fractal sample during mechanical loading and unloading. (d) Summary of the calculated and experimentally-measured stretchabilities for half-and-half samples along the x- and y-axes, delineating the transition point from elastic to plastic deformation.

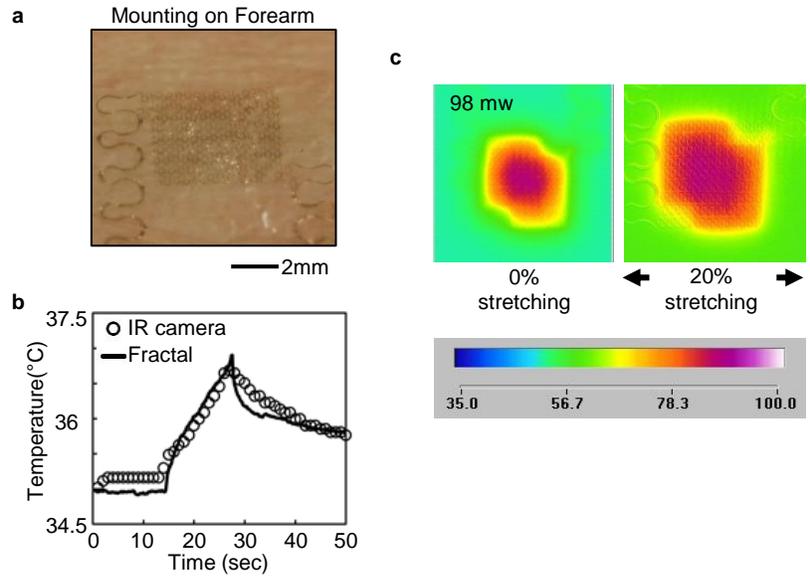


**Supplementary Figure 4. (a)** A comparison between experimental optical images and FEM-constructed images of an all-vertical second order structure under no strain, pre-strain, and stretching. **(b)** Plot of differential resistances measured in a sample for increasing amounts of maximum applied strain.

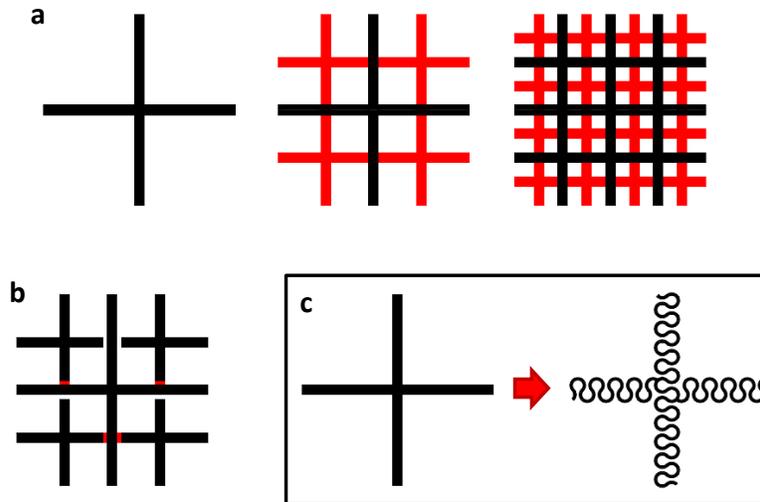
**Supplementary Figure 5.** Plot of silicon nanomembrane resistance as a function of tensile strain for two different half-and-half samples (black and red lines). Failure is clearly defined when the membranes crack and the resistance dramatically increases. The maximum elastic strains here are consistent with the simulated FEM strain of 63%.



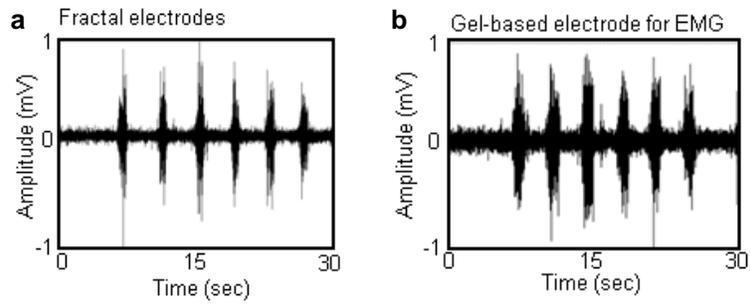
**Supplementary Figure 6. (a)** Calculated stretchability of an all-vertical Peano silicon nanomembrane strained along the y-axis for different membrane thicknesses. The planar geometry is identical to that presented in Figure 3. **(b)** As the membrane thickness increases, the strained membranes experience mechanical regimes spanning wrinkling, micro-buckling, and global buckling. The maximum device stretchability coincides with the micro-buckle regime, indicating that micro-buckles are particularly well suited for dissipating stress in the mechanically hard silicon membranes mounted on a soft substrate.



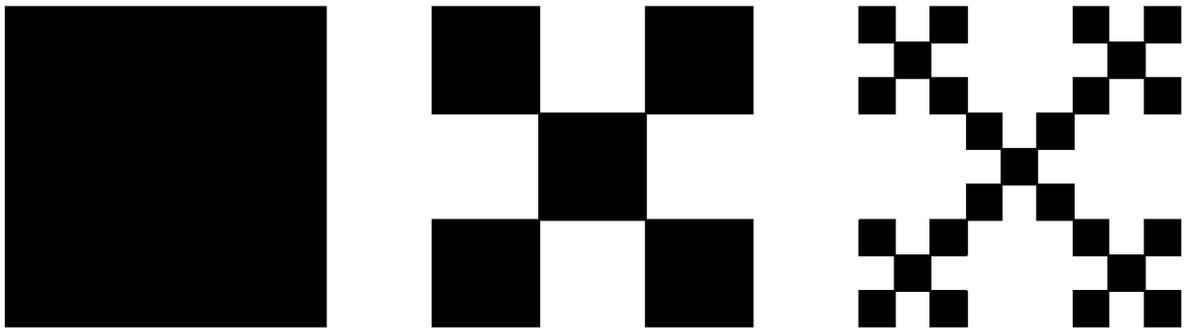
**Supplementary Figure 7. (a)** Optical image of a skin-mounted third order Peano temperature sensor. Simultaneous measurements of temperature on skin with an infrared camera **(b)** show good agreement. **(c)** A third order Peano device functions as a heater by passing larger amounts of current (order of 10mA) through the gold wires. The heat distribution within the device is uniform with and without stretching due to the space filling properties of the wire layout.



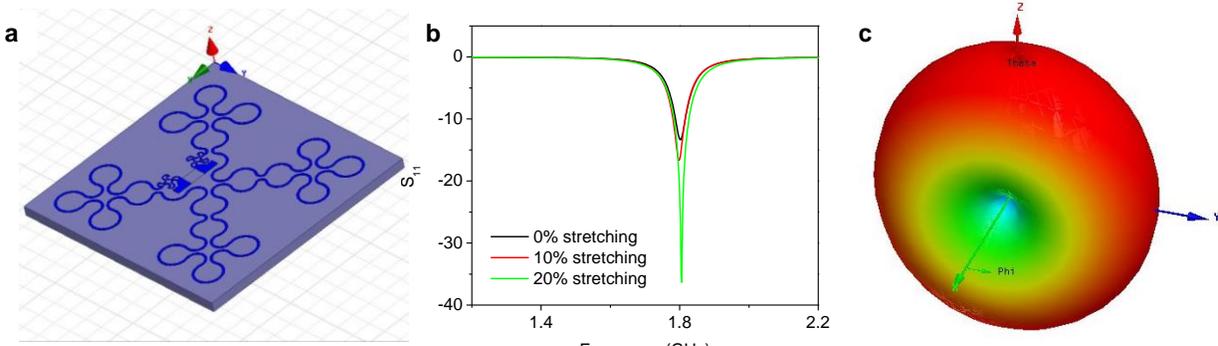
**Supplementary Figure 8.** (a) The first order structure is a cross. Higher order structures are iteratively constructed by adding crosses of exponentially smaller dimension, as shown. (b) For the electrodes featured here, the connectivity between the structures (red lines) is defined such that there are no closed loops of wire in the network. (c) The straight lines in the mathematical structure are replaced with serpentine layout to enable stretchability.



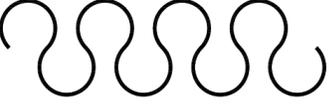
**Supplementary Figure 9.** A comparison of EMG signals taken from **(a)** fractal epidermal electrodes and **(b)** gel-based electrodes mounted on the forearm show comparable signal-to-noise ratios. The peaks are due to the clenching of the fist.



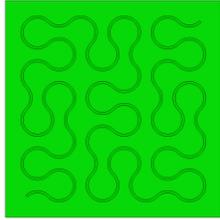
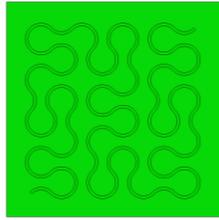
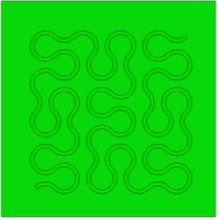
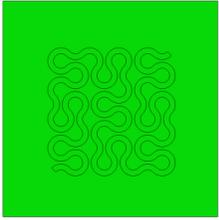
**Supplementary Figure 10.** Illustration of the first three iterations of the Vicsek fractal.



**Supplementary Figure 11.** (a) Box fractal antenna geometry as imported in HFSS. The feed point of the antenna is chosen so that the impedance of its fundamental mode sufficiently matches with the coaxial feed. (b) Return loss spectra of the box fractal antenna for three different levels of stretching. It is noted that the bandwidth of the box fractal is very narrow, due to the large inductive load built into the antenna. (c) Far field intensity profile of the unstretched antenna at resonance.

Serpentine Patterns	Arc Angle 	Elastic Stretchability
	270°	61%
	235°	60%
	180°	36%
	135°	18%
	90°	6%

**Supplementary Table 1.** Simulated uniaxial elastic stretchability for serpentine wires as a function of arc solid angle. The inset in the middle column defines the arc solid angle. The cross-sectional geometries and materials match those from Figure 3, and all structures have  $R = 620 \mu\text{m}$  and  $w = 70 \mu\text{m}$  (defined in Supplementary Figure 2). These simulations clearly display that elastic stretchability increases as a function of arc angle in these primitive serpentine geometries. As such, deterministically defining the arc section geometries in wire-type structures can help optimize the mechanics.

6.0 mm 	4.5 mm 	3.0 mm 	2.3 mm 	1.5 mm 
16%	28%	24%	23%	20%

**Supplementary Table 2.** Simulated biaxial stretchability as a function of unit cell size for half-and-half Peano structures. The cross-sectional geometries and materials match those from Figure 3, and all structures have  $w = 70 \mu\text{m}$  (defined in Supplementary Figure 2). The structures with unit cell sizes between 1.5 mm and 4.5 mm display biaxial stretchabilities greater than 20% and are compatible with the elastic properties of skin<sup>23</sup>.

## **Supplementary Note 1: Analysis of Fractal Geometries with MicroXCT**

Micro X-ray tomography (MicroXCT 400, Xradia) enables the imaging of the spatial topology of the various fractal structures. Experimental images in Figures 1 and 3 demonstrate the structural details of fractal patterns from the MicroXCT system. Two magnifications, 0.5 X and 4 X, provide the whole views and magnified views of the structures, respectively. Additional imaging parameters include a 40 KeV X-ray source and 200  $\mu$ A current, with 600 image frames recorded stepwise over 180 degrees. TXM Reconstructor software (Xradia) reconstruct the images, and the visualization software package (Amira 5.4.2, Visage Imaging) yields the “gray-scale” images in Figures 1 and 3.

## Supplementary Note 2: Mechanical Testing

The resistance of metal wires bonded to an elastomer is a function of both temperature and mechanical strain. The contribution of temperature is subtracted out to purely monitor changes in the metal resistance due to mechanical strain. The first step is to measure the resistance of the unstrained fractal samples as a function of temperature to obtain  $dR/dT$ , using a hot plate and IR camera (FLIR SC7650). A linear fit using the least squares method with six resistance points in the range of 30°C and 45°C yields this calibration. During the mechanical measurement itself, the temperature-adjusted resistance ( $R$ ) follows from the continuously measured temperature ( $T$ ) as:

$$R = R_{device} - \frac{dR}{dT}(T - T_0) \quad (1)$$

$R_{device}$  is the measured resistance of the device from the four point measurement and  $T_0$  is a constant reference temperature set before the start of the experiment. To measure the relative change in resistance as a function of cyclic straining, the resistance  $R_{initial}$  of the unstrained device at temperature  $T_0$  is recorded, the device is strained for one minute and then unstrained with the mechanical stretcher, and the subsequent resistance  $R_{cycle}$  is measured and temperature-corrected relative to  $T_0$  using equation (1). The relative change in resistance is calculated as  $(R_{initial} - R_{cycle})/ R_{initial}$ . Supplementary Figure 3 presents details of the experimental setup. Supplementary Figure 4 displays FEM images and the elastic-plastic transition point for a pre-strained all-vertical Peano structure.

### Supplementary Note 3: Temperature Sensor and Microheater Testing

Wires with Peano-based layouts have utility as the principal component in high-precision temperature sensors and heaters. Such sensors can be calibrated against an infrared (IR) camera (A655SC, FLIR, USA), as performed above for mechanical testing. The response of a third order half-and-half Peano sensor is plotted in Supplementary Figure 7. The noise of the analogue-digital (A/D) converter ( $V_{A/D}$ ) and the electrical noise ( $V_{noise}$ ) determine the precision of the fractal temperature sensor using the expressions:

$$T_{A/D} = \left(\frac{dV}{dT}\right)^{-1} V_{A/D} \quad (1)$$

$$T_{noise} = \left(\frac{dV}{dT}\right)^{-1} V_{noise} \quad (2)$$

where  $dV/dT$  is from the temperature calibration. The precision of the fractal temperature sensor is 0.022 °C. Mounting a device on a forearm and recording the temperature simultaneously with the sensor and an IR camera illustrates applicability to measurements on the skin.

The same device can be used as a precision element for Joule heating. Infrared images of a device under 0% and 20% uniaxial strain show that the heating is uniform across the area of the device, which is indicative of the space-filling nature of the fractal construct (Supplementary Figure 7c). One application involves wound recovery, for which the delivery of a controlled amount of heat to the wound vicinity leads to increased local blood flow and vasodilatation and ultimately expedited recovery. The ability for these devices to function as both temperature sensors and heaters enables the measurement of other quantities, such as thermal conductivity.

#### **Supplementary Note 4: Fractal Antenna Design and Simulations**

The box fractal antenna layout utilizes the two dimensional box fractal illustrated in Supplementary Figure 10. Here, five versions of the (N-1)th geometry scale down and connect together to construct the Nth iterative geometry. A wire tracing around the perimeter of the box fractal creates the antenna layout; the sharp corners are rounded to enhance the mechanics. The deformed antenna analysis with HFSS simulations requires three steps: the undeformed antenna geometries import into Abacus, they numerically stretch with a predetermined strain, and this resulting geometry imports into HFSS for simulation (Supplementary Figure 11). The frequencies and magnitudes of the calculated S11 parameters, and the dipolar far field profiles, are consistent with those measured experimentally.

## Supplementary Note 5: MRI Imaging

The spin echo images use the following parameters:

TR 2000

TE 25

Averages 5

Slices 25

FOV 140 x 140 mm

Thickness: 3 mm

Flip angle: 60 deg

Resol: 256

Partial fourier: 5/8

Bandwidth: 130 Hz/Px